

**BaLn<sub>2</sub>Se<sub>4</sub> (Ln = Er, Tm and Yb)**

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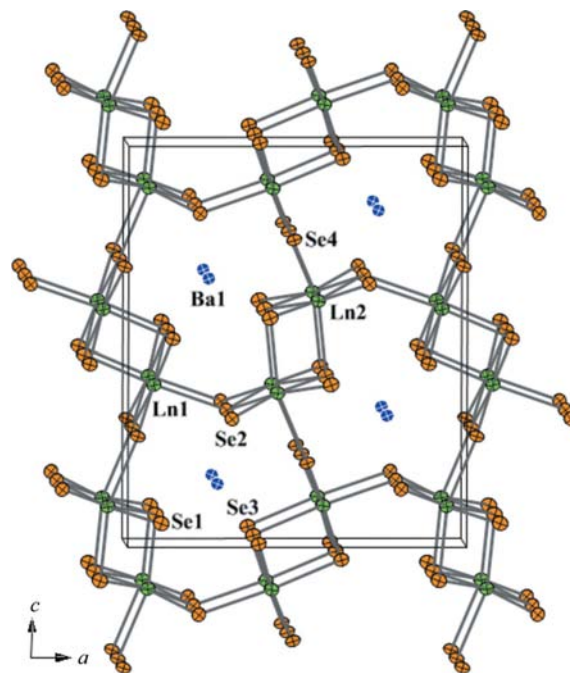
The compounds BaLn<sub>2</sub>Se<sub>4</sub> (Ln = rare-earth metal = lanthanide = Er, Tm and Yb), namely barium di(erbium/thulium/ytterbium) tetraselenide, crystallize in the orthorhombic space group *Pnma* in the CaFe<sub>2</sub>O<sub>4</sub> structure type. In this structure type, all atoms possess *m* symmetry. The Ln atoms are octahedrally coordinated by six Se atoms. A three-dimensional channel structure is formed by the corner- and edge-sharing of these LnSe<sub>6</sub> octahedra. The Ba atoms are coordinated to eight Se atoms in a bicapped trigonal–prismatic arrangement, and they occupy the channels of the three-dimensional framework.

**Comment**

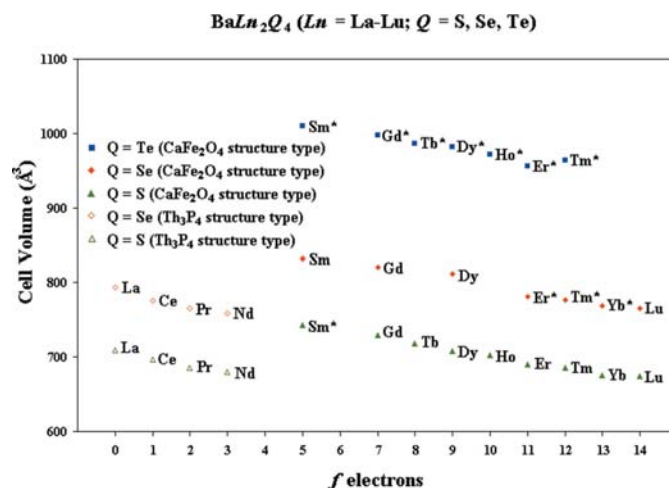
Ternary compounds of the family ALn<sub>2</sub>Q<sub>4</sub> (A = alkaline earth, Ln = lanthanide, and Q = S, Se and Te) have been extensively studied because of their magnetic and optical properties. Additionally, these compounds present intriguing structural variations. Members of this family crystallize in four structure types, namely MgAl<sub>2</sub>O<sub>4</sub> (Passerini, 1930), Th<sub>3</sub>P<sub>4</sub> (Meisel, 1939), Yb<sub>3</sub>S<sub>4</sub> (Chevalier *et al.*, 1967) and CaFe<sub>2</sub>O<sub>4</sub> (Decker & Kasper, 1957). The ALn<sub>2</sub>Q<sub>4</sub> compounds in the MgAl<sub>2</sub>O<sub>4</sub> structure type have A replacing Mg and Ln replacing Al. Those in the Th<sub>3</sub>P<sub>4</sub> structure type have the A and Ln cations disordered over the Th site in a 1:2 ratio. Those in the Yb<sub>3</sub>S<sub>4</sub> structure type have the A cation replacing the one Yb<sup>2+</sup> site and the Ln cations replacing the two Yb<sup>3+</sup> sites. The ALn<sub>2</sub>Q<sub>4</sub> compounds in the CaFe<sub>2</sub>O<sub>4</sub> structure type have A replacing Ca and Ln replacing Fe.

The members of the BaLn<sub>2</sub>Q<sub>4</sub> (Q = S, Se and Te) subfamily whose structures are known adopt either the Th<sub>3</sub>P<sub>4</sub> or the CaFe<sub>2</sub>O<sub>4</sub> structure type. The present compounds, *viz.* BaLn<sub>2</sub>Se<sub>4</sub> (Ln = Er, Tm and Yb), crystallize in the orthorhombic space group *Pnma* in the CaFe<sub>2</sub>O<sub>4</sub> structure type. Fig. 1 shows the unit cell of these BaLn<sub>2</sub>Se<sub>4</sub> compounds. The formal oxidation states may be assigned as Ba<sup>2+</sup>, Ln<sup>3+</sup> and Se<sup>2-</sup>. The two crystallographically unique Ln atoms are six-coordinated, bonded to an octahedral arrangement of Se atoms. The Ba atom is surrounded by eight Se atoms in a

bicapped trigonal–prismatic geometry. Each LnSe<sub>6</sub> octahedron shares edges with a neighboring octahedron to form two types of dimers, namely Ln(1)<sub>2</sub>Se<sub>10</sub> and Ln(2)<sub>2</sub>Se<sub>10</sub>. These dimers share edges in the [010] direction to form infinite chains. Additionally, in the (010) plane each Ln(1)<sub>2</sub>Se<sub>10</sub> dimer shares four corners with Ln(2)<sub>2</sub>Se<sub>10</sub> dimers, and *vice versa*, to form extended sheets. The combination of chains in the [010] direction and sheets in the (010) plane forms a three-dimen-



**Figure 1**  
A view down [010] of the unit cell of BaLn<sub>2</sub>Se<sub>4</sub> (Ln = Er, Tm and Yb), with displacement ellipsoids at the 95% probability level.



**Figure 2**  
A plot of cell volume *versus* the number of *f* electrons for the BaLn<sub>2</sub>Q<sub>4</sub> subfamily. An asterisk indicates a result from single-crystal X-ray diffraction data; otherwise, the result is from powder X-ray diffraction data.

sional channel structure, with Ba<sup>2+</sup> cations residing in the channels.

Fig. 2 shows the relation between the cell volume and lanthanide element (number of *f* electrons) for the present and previously known members of the BaLn<sub>2</sub>Q<sub>4</sub> subfamily (Patrie *et al.*, 1964; Flahaut *et al.*, 1965; Carpenter & Hwu, 1992a; Narducci *et al.*, 2000). The trend in cell volume *versus* number of *f* electrons approximately follows the lanthanide contraction. The earlier lanthanides of Te could not be synthesized (Narducci *et al.*, 2000). The later lanthanides of all three chalcogens crystallize in the CaFe<sub>2</sub>O<sub>4</sub> structure type, whereas the earlier lanthanides of Se and S crystallize in the Th<sub>3</sub>P<sub>4</sub> structure type. It is not clear why there is this change from the lower-density CaFe<sub>2</sub>O<sub>4</sub> structure type to the higher-density Th<sub>3</sub>P<sub>4</sub> structure type between Nd<sup>3+</sup> and Sm<sup>3+</sup>.

Nonstoichiometry has been claimed in some of the ALn<sub>2</sub>Q<sub>4</sub> compounds on the basis of single-crystal X-ray diffraction data. For example, Ba<sub>1-x</sub>Sm<sub>2</sub>S<sub>4-x</sub> (*x* = 0.10; Carpenter & Hwu, 1992a) is said to exhibit deficiencies on both the alkaline-earth and the chalcogen atoms. Additionally, Ca<sub>1-x</sub>Yb<sub>2+x</sub>S<sub>4</sub> (*x* = 0.04; Carpenter & Hwu, 1992b) of the Yb<sub>3</sub>S<sub>4</sub> structure type is said to show a slight deficiency of Ca, which is compensated for by a mixing of Yb<sup>2+</sup> on the Ca<sup>2+</sup> site. Such mixing is possible because the ionic radii for seven-coordinate Ca<sup>2+</sup> and Yb<sup>2+</sup> are approximately equal (1.06 and 1.08 Å, respectively; Shannon, 1976). Refinement of the present BaLn<sub>2</sub>Se<sub>4</sub> compounds in which the occupancy factors were allowed to vary revealed no evidence either of alkaline-earth or chalcogen deficiencies or of Ba<sup>2+</sup> and Yb<sup>2+</sup> disorder. Mixing of Ba<sup>2+</sup> and Yb<sup>2+</sup> on the eight-coordinate alkaline-earth site is disfavored because of the mismatch in ionic radii (1.42 and 1.14 Å, respectively; Shannon, 1976). In the absence of support from other physical measurements, such claims, including the present one, based solely on X-ray diffraction data, must be viewed as provisional.

## Experimental

The BaLn<sub>2</sub>Se<sub>4</sub> compounds were synthesized by high-temperature solid-state reactions. The following reagents were used as obtained: Er (Alfa Aesar, 99.5%), Tm (Strem Chemicals, 99.9%), Yb (Alfa Aesar, 99.99%), filings from a Ba rod (Johnson Matthey, 99.5%), Se (Cerac, 99.999%) and Sb (Aldrich, 99.5%). The BaLn<sub>2</sub>Se<sub>4</sub> compounds were synthesized from Ln (0.13 mmol), Ba (0.12 mmol) and Se (0.76 mmol). Sb (0.12 mmol) was added as a flux to aid in crystallization of the final product. The reactants were loaded into fused-silica tubes under an argon atmosphere in a glove-box. The tubes were evacuated to 10<sup>-4</sup> Torr, flame sealed and then placed in a computer-controlled furnace. The tubes were heated to 1123 K over a period of 17 h, held at 1123 K for 6 d, cooled at a rate of 3 K h<sup>-1</sup> to 673 K and annealed at 673 K for one week, and finally the furnace was turned off. The resulting thin orange (Er and Tm) and red (Yb) needles were manually extracted from the product. The yield was approximately 30% based on Ln. The product also contained a gray metallic melt, as well as a black powder. Analysis of the crystals with an EDX-equipped (energy-dispersive X-ray spectroscopy) Hitachi S-3400 scanning electron microscope showed the presence of Ba, Ln and Se, but not of Sb.

## Compound (I)

### Crystal data

BaEr <sub>2</sub> Se <sub>4</sub>	<i>V</i> = 781.7 (3) Å <sup>3</sup>
<i>M<sub>r</sub></i> = 787.70	<i>Z</i> = 4
Orthorhombic, <i>Pnma</i>	Mo <i>K</i> α radiation
<i>a</i> = 12.628 (3) Å	<i>μ</i> = 44.77 mm <sup>-1</sup>
<i>b</i> = 4.1398 (8) Å	<i>T</i> = 100 K
<i>c</i> = 14.953 (3) Å	0.30 × 0.05 × 0.05 mm

### Data collection

Bruker APEXII diffractometer	8227 measured reflections
Absorption correction: numerical	1036 independent reflections
(SADABS; Sheldrick, 2008b)	987 reflections with <i>I</i> > 2σ( <i>I</i> )
<i>T<sub>min</sub></i> = 0.025, <i>T<sub>max</sub></i> = 0.196	<i>R<sub>int</sub></i> = 0.034

### Refinement

<i>R</i> [ <i>F</i> <sup>2</sup> > 2σ( <i>F</i> <sup>2</sup> )] = 0.025	44 parameters
<i>wR</i> ( <i>F</i> <sup>2</sup> ) = 0.066	Δ <i>ρ</i> <sub>max</sub> = 1.59 e Å <sup>-3</sup>
<i>S</i> = 1.54	Δ <i>ρ</i> <sub>min</sub> = -2.68 e Å <sup>-3</sup>
1036 reflections	

## Compound (II)

### Crystal data

BaTm <sub>2</sub> Se <sub>4</sub>	<i>V</i> = 776.9 (3) Å <sup>3</sup>
<i>M<sub>r</sub></i> = 791.04	<i>Z</i> = 4
Orthorhombic, <i>Pnma</i>	Mo <i>K</i> α radiation
<i>a</i> = 12.605 (3) Å	<i>μ</i> = 46.28 mm <sup>-1</sup>
<i>b</i> = 4.1311 (8) Å	<i>T</i> = 110 K
<i>c</i> = 14.919 (3) Å	0.22 × 0.06 × 0.05 mm

### Data collection

Bruker APEXII diffractometer	8761 measured reflections
Absorption correction: numerical	1070 independent reflections
(SADABS; Sheldrick, 2008b)	1041 reflections with <i>I</i> > 2σ( <i>I</i> )
<i>T<sub>min</sub></i> = 0.021, <i>T<sub>max</sub></i> = 0.178	<i>R<sub>int</sub></i> = 0.035

### Refinement

<i>R</i> [ <i>F</i> <sup>2</sup> > 2σ( <i>F</i> <sup>2</sup> )] = 0.018	44 parameters
<i>wR</i> ( <i>F</i> <sup>2</sup> ) = 0.042	Δ <i>ρ</i> <sub>max</sub> = 1.69 e Å <sup>-3</sup>
<i>S</i> = 1.27	Δ <i>ρ</i> <sub>min</sub> = -1.36 e Å <sup>-3</sup>
1070 reflections	

## Compound (III)

### Crystal data

BaYb <sub>2</sub> Se <sub>4</sub>	<i>V</i> = 769.2 (3) Å <sup>3</sup>
<i>M<sub>r</sub></i> = 799.26	<i>Z</i> = 4
Orthorhombic, <i>Pnma</i>	Mo <i>K</i> α radiation
<i>a</i> = 12.580 (3) Å	<i>μ</i> = 47.99 mm <sup>-1</sup>
<i>b</i> = 4.1150 (8) Å	<i>T</i> = 100 K
<i>c</i> = 14.860 (3) Å	0.17 × 0.02 × 0.02 mm

### Data collection

Bruker APEXII diffractometer	8818 measured reflections
Absorption correction: numerical	1056 independent reflections
(SADABS; Sheldrick, 2008b)	1012 reflections with <i>I</i> > 2σ( <i>I</i> )
<i>T<sub>min</sub></i> = 0.113, <i>T<sub>max</sub></i> = 0.429	<i>R<sub>int</sub></i> = 0.034

### Refinement

<i>R</i> [ <i>F</i> <sup>2</sup> > 2σ( <i>F</i> <sup>2</sup> )] = 0.018	44 parameters
<i>wR</i> ( <i>F</i> <sup>2</sup> ) = 0.042	Δ <i>ρ</i> <sub>max</sub> = 1.35 e Å <sup>-3</sup>
<i>S</i> = 1.22	Δ <i>ρ</i> <sub>min</sub> = -1.90 e Å <sup>-3</sup>
1056 reflections	

The structures were standardized by means of the program *STRUCTURE TIDY* (Gelato & Parthé, 1987).

For all compounds, data collection: *APEX2* (Bruker, 2008); cell refinement: *APEX2*; data reduction: *SAINT* (Bruker, 2008); program(s) used to solve structure: *SHELXS* (Sheldrick, 2008a); program(s) used to refine structure: *SHELXL* (Sheldrick, 2008a); molecular graphics: *CrystalMaker* (Palmer, 2009); software used to prepare material for publication: *SHELXTL* (Sheldrick, 2008a).

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Supplementary data for this paper are available from the IUCr electronic archives (Reference: SQ3208). Services for accessing these data are described at the back of the journal.

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## References

Bruker (2008). *APEX2* (Version 2008.6-1) and *SAINT* (Version 7.34a). Bruker AXS Inc., Madison, Wisconsin, USA.

- Carpenter, J. D. & Hwu, S.-J. (1992a). *Acta Cryst.* **C48**, 1164–1167.  
Carpenter, J. D. & Hwu, S.-J. (1992b). *J. Solid State Chem.* **97**, 332–339.  
Chevalier, R., Laruelle, P. & Flahaut, J. (1967). *Bull. Soc. Fr. Mineral. Cristallogr.* **90**, 564–574.  
Decker, B. F. & Kasper, J. S. (1957). *Acta Cryst.* **10**, 332–337.  
Flahaut, J., Guittard, M., Patrie, M., Pardo, M. P., Golabi, S. M. & Domange, L. (1965). *Acta Cryst.* **19**, 14–19.  
Gelato, L. M. & Parthé, E. (1987). *J. Appl. Cryst.* **20**, 139–143.  
Meisel, K. (1939). *Z. Anorg. Allg. Chem.* **240**, 300–312.  
Narducci, A. A., Yang, Y., Digman, M. A., Sipes, A. B. & Ibers, J. A. (2000). *J. Alloys Compd.* **303–304**, 432–439.  
Palmer, D. (2009). *CrystalMaker*. Version 2.1.5. CrystalMaker Software Ltd, Yarnton, Oxfordshire, England.  
Passerini, L. (1930). *Gazz. Chim. Ital.* **60**, 389–399.  
Patrie, M., Golabi, S. M., Flahaut, J. & Domange, L. (1964). *C. R. Acad. Sci.* **259**, 4039–4042.  
Shannon, R. D. (1976). *Acta Cryst.* **A32**, 751–767.  
Sheldrick, G. M. (2008a). *Acta Cryst.* **A64**, 112–122.  
Sheldrick, G. M. (2008b). *SADABS*. Version 2008/1. University of Göttingen, Germany.